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Performance Study of NiO-TiO₂-CuO Nanocomposite Supported by Reduced Graphene Oxide as an Anode Candidate for Lithium-Ion Battery Development

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Abstract

In an effort to enhance the performance of lithium-ion batteries (LIBs), this study developed a NiO-TiO₂-CuO nanocomposite supported by reduced graphene oxide (rGO) as an anode material. The nanocomposite was synthesized via a hydrothermal method and characterized using FTIR, XRD, and SEM-EDX techniques to understand its structure and material properties. The FTIR spectrum confirmed the presence of C=C bonds (1612-1512 cm⁻¹) and C-O bonds (1147-1099 cm⁻¹) from rGO, as well as Ni-O (408 cm⁻¹), Cu-O (669 cm⁻¹), and Ti-O (549 cm⁻¹). The XRD patterns revealed the crystalline phases of NiO at $2\theta = 37^{\circ}$ (111), 43° (200), and 62.8° (200); TiO₂ at $2\theta = 25.3^{\circ}$ (101), 48° (101), and 55° (211); and Cu-O at 2θ = 35.6° (111) and 39.8° (022). SEM-EDX images showed small aggregated particles forming a relatively uneven surface with spherical morphology, with an average particle size of 33.25 nm. Electrochemical testing using cyclic voltammetry (CV) demonstrated that the material exhibited a stable specific capacity (C_{sp}) of 6.3 mAh/g after five cycles at a scan rate of 1 V/s. Additionally, the specific capacity significantly increased to 44.15 mAh/g at a scan rate of 0.05 V/s, indicating excellent electrochemical performance. These results suggest that the NiO-TiO₂-CuO/rGO nanocomposite has potential as an efficient anode material for lithium-ion battery applications, offering good cycle stability and enhanced energy storage capacity.

1. Introduction

In the era of global energy transition, batteries have become a crucial component to support sustainable energy needs. Batteries require materials with high energy capacity and fast charge-discharge efficiency. Among the types of batteries currently being developed and widely applied in daily life, lithium-ion batteries (LIBs) stand out due to their capability as high-capacity energy storage devices, excellent cycle performance, and environmental friendliness. However, LIBs face challenges such as imbalance and relatively short battery lifespan [1]. To enhance battery performance and capacity, intensive research is being conducted to develop more advanced electrode materials. One promising approach is using transition metal oxide (TMO) composites. This combination offers the potential to improve energy storage capacity, optimize charge rates, and extend battery cycle life [2, 3, 4, 5, 6]. This research focuses on developing anode materials to improve capacity, cycle performance, and battery lifespan.

TMO materials such as NiO, CuO, and TiO₂ were selected for this study due to their extensively researched high capacities [7, 8]. NiO is recognized as a potential anode material because of its high theoretical capacity





and stable electrochemical properties. Even at low current densities, NiO demonstrates good reversible capacity and cycle stability [9]. Additionally, NiO exhibits high thermal stability, enabling it to withstand high-temperature and high-voltage battery operating conditions. Several studies have shown that NiO has sufficient ionic conductivity, making it an excellent conductive material for faster electron transfer [10, 11]. Another unique feature of NiO is its nanostructure, which allows it to be easily combined with other conductive materials. It has been reported that CuO coated with a thin conformal layer of NiO as an anode material exhibits excellent electrochemical performance. The presence of the NiO coating effectively facilitates ion diffusion into CuO, maintains the advantages of high surface area, and enhances cycle performance, leading to improved battery storage capacity [11].

In this study, TiO₂ was selected as a Ni-based electrode modification material based on literature reviews. Although progress has been made in the energy storage properties, capacity enhancement, and cycle stability of these materials, challenges remain for researchers. TiO₂ is considered a potential alternative anode material to replace commercially available graphite. Its advantages include preventing dendrite growth, low volume change, environmental friendliness, chemical stability, and low-cost [12, 13, 14, 15]. In electrochemical systems, TiO₂ holds significant promise for energy storage devices due to its fast electron transfer and reduced diffusion distance [15]. Furthermore, TiO₂ offers a high theoretical capacity (332 mAh g^{-1}) [16], but its low electronic conductivity limits its storage capacity. Therefore, developing TiO₂ with short diffusion lengths for electron and ion transport is essential. Various strategies have been explored to enhance TiO₂ capacity. Previous reports have shown that unique TiO₂@NiO structures deliver high capacity and cycle stability [17].

Conventional LIB anodes are typically composed of graphite. However, graphite can only operate at low potentials, limiting its theoretical capacity to meet the demands of modern electronics [18, 19]. To address this issue, we innovated by modifying graphite into reduced graphene oxide (rGO). Our investigation shows that rGO has a larger surface area than graphite [20], which can enhance battery capacity, cycling life, and charge-discharge efficiency [18]. Our studies also indicate that rGO combined with TiO₂ exhibits high energy storage and

excellent electrochemical performance due to the wellorganized three-phase composite electrode structure, ensuring stable electron transfer. This new structure reduces crystal expansion during the discharge process, enhances ion diffusion rates, and shortens electron transmission distances [18, 21].

Several previous studies have documented various comparisons involving different anode materials used in LIBs. These studies have examined various highperformance anodes and their electrochemical performance, with the findings compiled and presented in Table 1.

Based on our investigation, referring to Table 1 and several reviewed studies, the presence of TMO modifications has been shown to effectively enhance capacity, cycling life, and current stability. This research aims to evaluate the electrochemical performance of NiO-TiO₂-CuO nanocomposites supported by rGO as an energy storage electrode material for batteries. Material characterization using techniques such as XRD, SEM-EDX, and FTIR will be conducted to understand the structure and properties of the material. Additionally, electrochemical testing, including CV, will be used to assess the energy storage performance of the nanocomposite. Thus, this study is expected to contribute to developing more efficient and sustainable LIB materials.

2. Experimental

2.1. Materials

The materials used in this study were titanium dioxide (TiO₂) Degussa, ethanol deionized water, potassium permanganate (KMnO₄), hydrogen peroxide (H₂O₂), sulfuric acid (H₂SO₄), graphite, nickel sulfate hexahydrate (NiSO₄.6H₂O), sodium hydroxide (NaOH), copper sulfate pentahydrate (CuSO₄.5H₂O), sodium nitrate (NaNO₃), and potassium ferricyanide K₃[Fe(CN)₆].

2.2. Synthesis of TiO₂

 TiO_2 nanoparticles were prepared using the annealing method. Initially, 10 g of TiO_2 Degussa was dissolved in a mixture of deionized water and ethanol (1:1). The suspension was then sonicated for 1 hour at 80°C. After sonication, the suspension was filtered and dried at 500°C in a sealed container for 3 hours [22, 23, 24, 25].

Material	C _{sp} (mAh/g)	Cycle	Reference
TiO ₂	332	100	[16]
TiO ₂ /Ni/C nanohybrids	422	100	[26]
NiO/TiO ₂	152.4	100	[27]
TiO ₂ /rGO (few-layers)	344.3	100	[28]
Ni-TiO ₂ /Graphene	283	200	[29]
TiO ₂ /NiO/rGO	245	200	[18]
NGr@NiO/TiO2 hollow nanospheres	839.84	-	[30]
NiO-TiO ₂ -CuO/rGO nanocomposite	This research		

Table 1. Research developments related to anodes



Figure 1. Research scheme of synthesis procedure for the NiO-TiO₂-CuO/rGO electrode

2.3. Synthesis of rGO

Graphene nanosheets incorporating rGO were synthesized using a modified Hummers' method [31, 32, 33]. Graphite powder, KMnO₄ (99%), H₂O₂ (30%), and H₂SO₄ (95%) were used. First, 25 mL of concentrated sulfuric acid was mixed with 1 g of graphite powder, and 3.5 g of KMnO₄ was slowly added to the solution. The mixture was stirred at 35°C for 2 hours, followed by the addition of 40 mL of deionized water. After 1 hour of stirring, hydrogen peroxide was added dropwise until the residual H₂O₂ gas disappeared. Graphene oxide was collected via centrifugation and dried at 70°C for 12 hours. The prepared graphene oxide was reduced using a solidstate microwave irradiation method. Briefly, graphene oxide powder (90 wt.%) was mixed with graphene nanosheet powder (10 wt.%), which acted as an effective microwave susceptor to produce high-quality rGO. The mixture was transferred to a quartz tube and reduced by microwave irradiation using a microwave oven. The microwave treatment was performed at 1600 W in pulsed irradiation mode [34].

2.4. Synthesis of NiO

NiO nanoparticles were synthesized using a precipitation method based on the literature [35, 36, 37]. A 250 mL solution of 1.2 M NaOH was slowly added dropwise into a 250 mL solution of 0.5 M NiSO₄.6H₂O while stirring at 1000 rpm using a magnetic stirrer at room temperature. The resulting Ni(OH)₂ precipitate was separated and washed until free of sulfate ions, and the filtrate reached a neutral pH. The cleaned Ni(OH)₂ solid was dried in an oven at 70°C for 24 hours. The dried solid was then calcined at 800°C for 2 hours.

2.5. Synthesis of CuO

CuO nanoparticles were synthesized using a simple wet chemical method based on the literature [38, 39]. First, a 250 mL solution of 0.5 M CuSO₄.5H₂O was prepared, and a 250 mL solution of 1.2 M NaOH was added dropwise using a burette while stirring at 1000 rpm with a magnetic stirrer at room temperature. The resulting Cu(OH)₂ precipitate was separated via filtration and washed with distilled water until free of sulfate ions, and the filtrate reached a neutral pH. The washed precipitate was dried in an oven at 100°C for 3 hours. The dried solid

was then calcined at 400°C for 2 hours to obtain wellcrystallized CuO nanoparticles.

2.6. Characterization

The electrode morphology was characterized using X-ray diffraction (XRD) (Shimadzu 6000) at $2\theta = 20-80^{\circ}$ with Cu-K α radiation ($\lambda = 1.54060$ Å). The nanocomposite morphology and atomic composition were analyzed using Scanning Electron Microscopy-Energy Dispersive X-ray spectroscopy (SEM-EDX) (HITACHI SU3500). Chemical bonds and functional groups were identified using Fourier Transform Infrared Spectroscopy (FT-IR) (Shimadzu Varian 4300 spectrophotometer). The electrochemical properties of the nanocomposite were characterized using cyclic voltammetry with a DY2100 potentiostat.

2.7. Preparation of NiO-TiO₂-CuO/rGO Electrode

The NiO-TiO₂-CuO/rGO composite was prepared with a mass ratio of NiO, TiO₂, CuO, and rGO at 10:30:10:50 (w/w%). The composite was dissolved in 50 mL of deionized water and sonicated for 1 hour. The suspension was then transferred to a 100 mL Teflon-lined stainless steel autoclave and heated at 200°C for 6 hours. After the reaction, the autoclave was cooled to room temperature, and the product was washed several times with deionized water and ethanol, followed by drying at 60°C for 24 hours. The product was further hydrothermally treated at 350°C for 2 hours. For electrode preparation, 0.05 g of the NiO-TiO₂-CuO/rGO nanocomposite was mixed with 0.3 g of paraffin oil and stirred at 400 rpm for 15 minutes while heating at 80°C. Finally, the NiO-TiO₂-CuO/rGO paste was packed into a 3 mm diameter glass tube, pressed gently, smoothed on the surface, and connected with a copper wire.

2.8. Electrochemical Performance Testing of NiO-TiO₂-CuO/rGO Electrode

The electrochemical properties of the NiO-TiO₂-CuO/rGO electrode were evaluated using cyclic voltammetry (CV) at scan rates of 1, 0.5, 0.2, 0.1, and 0.05 V/s within a potential range of -0.8 V to 0.8 V. The tests were conducted in an electrochemical cell containing 0.01 M K₃[Fe(CN)₆] as the analyte and 0.1 M NaNO₃ as the electrolyte.

3. Results and Discussion

3.1. Morphological and Structure of NiO-TiO₂-CuO/rGO Nanocomposite

In this study, NiO-TiO₂-CuO/rGO was obtained by combining NiO, TiO₂, CuO, and rGO (Figure 1). The success of the research method was confirmed through various analytical techniques. FTIR analysis (Figure 2) and a literature review (Table 2) identify the functional groups corresponding to each spectral peak. Figure 2a shows a peak around 3400-3200 cm⁻¹ (3219 cm⁻¹), indicating the presence of O-H stretching, which may originate from adsorbed water molecules on the nanocomposite surface [6] or oxygen-containing groups on the rGO surface. The peak around 2439-2312 cm⁻¹ corresponds to C-H stretching, likely derived from organic solvents [40]. In this context, this band may indicate residual ethanolbased organic compounds during the TiO₂ synthesis process. Additionally, the peak at approximately 1147-1099 cm⁻¹ corresponds to C-O stretching vibrations [41, 42], confirming the presence of oxygen in rGO or TMOs.

Figure 2b shows a small peak around $1612-1512 \text{ cm}^{-1}$, which can be attributed to the asymmetric stretching of C=C bonds in graphene sheet vibrations [43, 44]. The peaks around 669 cm⁻¹ and 549 cm⁻¹ are characteristic of the stretching vibrations of metal-oxygen bonds, specifically Cu-O and Ni-O [40, 45, 46]. The peak around 408 cm⁻¹ is likely related to the stretching vibration of the Ti-O group in the nanocomposite structure [47].

The XRD pattern shows several intensity peaks that can be identified to determine the crystalline phases present in the nanocomposite. The XRD diffraction pattern of NiO-TiO₂-CuO/rGO is displayed in Figure 3. The dominant peaks observed between $2\theta = 20^{\circ}$ and 80° indicate the presence of various crystalline phases. Strong peaks around $2\theta = 37^{\circ}$, 43° , and 62.8° were observed and can be indexed to the Miller indices (111), (200), and (220), which are characteristic of the NiO structure (ref. JCPDS 45-1049) [41, 42].



Figure 2. FTIR spectra of (a) NiO-TiO₂-CuO/rGO nanocomposite and (b) C=C, C-O, Ti-O, Cu-O and Ni-O

Table 2. Reference FTIR data for NiO-TiO₂-CuO/rGO

Functional groups	Reference wavenumber (cm ⁻¹)	Wavenumber (cm ⁻¹)	Reference
Ti-O	800-400	408	[45, 46]
Cu-O	601	669	[40]
Ni-O	549	550	[48]
C=C	1493–1630	1612-1512	[49]
C-0	1065-1226	1099	[49]



Figure 3. XRD pattern of NiO-TiO₂-CuO/rGO

Peaks observed around $2\theta = 25.3^{\circ}$, 48° , and 55° were indexed to the Miller indices (101), (101), and (211), likely corresponding to TiO₂ in the anatase phase, which is commonly found in TiO₂ materials (ref. JCPDS 21-1272) [50]. Low-intensity peaks around $2\theta = 35.6^{\circ}$ and 39.8° were observed and can be indexed to the Miller indices (111) and (022), which may be associated with the CuO phase (ref. JCPDS 05-0661) [51]. A weak peak around 23.9° was observed and indexed to the Miller index (002), related to conjugated rGO [49]. Thus, the analyzed XRD diffraction pattern demonstrates the successful synthesis of NiO-TiO₂-CuO/rGO, and these findings are supported by EDX characterization data (Figure 4c).

To confirm the findings from the XRD analysis, SEM-EDX characterization was also conducted. Figure 4a provides a a thin-layer composite structure with welldispersed nanocomposite particles. The observed structure shows small-sized particles aggregated together, forming a relatively uneven surface. Spherical nanoparticles appear to cover almost the entire layer surface, indicating good particle dispersion. Pore structures or small gaps are visible in some areas, supporting the porous nature of the material, as seen in Figure 4b. This is consistent with the morphology of the nanocomposite, where each component (NiO, TiO₂, CuO, and rGO) is integrated into a single structure [52, 53]. The NiO-TiO₂-CuO/rGO nanocomposite, subjected to hightemperature treatment, significantly influences particle size and minor changes in its structural shape.

The presence of each component (NiO, TiO₂, CuO, and rGO) in the material was further confirmed using EDX, as shown in Figure 4c. The results indicate the concentrations of NiO, TiO₂, CuO, and rGO to be 1.22%, 31.93%, 1.46%, and 4.12%, respectively. The inconsistent ratio likely results from non-homogeneous mixing, causing variations in composition across different regions. Since EDX analyzes specific points, it may not represent the overall material distribution. Additionally, during the heating process in the synthesis, some elements may have undergone segregation or changes in distribution. rGO may be more prone to burning or degradation than TiO₂ and CuO, which could explain its lower percentage in the EDX results than expected [54].



Figure 4. Morphology and composition of NiO-TiO₂-CuO/rGO: (a) 20,000× magnification, (b) 30,000× magnification, (c) Elemental composition, (d) Particle size distribution

To determine the particle size distribution of NiO-TiO₂-CuO/rGO, the SEM characterization results were processed using ImageJ software with its particle analysis feature. The particle diameter distribution ranges from 20 to 55 nm, with an average diameter of 33.25 nm and an R- squared value of 0.94568, indicating good data quality (Figure 4d). The consistency in particle size demonstrates that the synthesis method successfully produced a relatively uniform size distribution, confirming that the optimized material properties can enhance electrochemical performance.

3.2. Electrochemical Performace

The electrochemical performance of the NiO-TiO₂-CuO/rGO nanocomposite was tested using the CV (cyclic voltammetry) method. The testing was conducted using platinum as the counter electrode, Ag/AgCl as the reference electrode, and NiO-TiO₂-CuO/rGO as the working electrode. CV of the working electrode was performed at scan rates ranging from 5 to 100 mV/s within a voltage range of -0.8 to 0.8 V, as shown in Figures 5a and 5b.



Figure 5. The voltammogram of NiO-TiO₂-CuO/rGO (a) first-five cycles at a scan rate of 1 V/s, (b) Cyclic variation, (c) C_{sp}: first-five cycles at a scan rate of 1 V/s, (d) C_{sp}: Cyclic variation

Determining anode stability requires multiple measurements. The formation of oxidation and reduction peaks and stable voltammograms allows for assessing anode reproducibility and reversibility. The curves in Figure 5a demonstrate the stable performance of the nanocomposite during the first to fifth CV cycles. This good stability is evidenced by the R values for the oxidation and reduction processes, which are 0.97984 and 0.97697, respectively, indicating high data quality. Minor changes in current intensity during the initial cycles suggest that the material undergoes an electrochemical activation process at the beginning of the cycles. This is a common phenomenon in the initial cycles, where the anode material adapts to store and release ions [55].

The curve in Figure 5b shows improved performance with each cycle, forming a well-defined voltammogram. The voltammogram indicates an increase in the area under the curve as the scan rate increases [56]. Each cycle exhibits a cathodic peak at ~0.18 V with an R-value of 0.98695 during discharge, followed by an anodic peak at ~-0.2 V with an R-value of 0.98023. R-values greater than 0.9 confirm the high quality of the data. This behavior suggests that the nanocomposite facilitates easier electron insertion and extraction processes, indicating that NiO-TiO₂-CuO/rGO has potential as an anode material with excellent electrochemical properties.

The graphs in Figures 5c and 5d show the capacity of $NiO-TiO_2-CuO/rGO$ based on the voltammogram curves generated in Figures 5a and 5b. The calculation process follows Equation (1).

$$Q = \int_{V_1}^{V_2} I(V) dV \tag{1}$$

Plot the current (I) against the potential (V) and identify the relevant area under the curve (reduction or oxidation region). Then, integrate the current with respect to the potential to obtain the charge (Q). The specific capacity (C_{sp}) is then calculated using Equation (2).

$$C_{sp} = \frac{Q(C)}{3.6 \times m} \tag{2}$$

Where m is the mass (in grams), and the factor 3.6 converts Coulombs to milliampere-hours (mAh).

The curve in Figure 5c shows a decrease in C_{sp} for NiO-TiO₂-CuO/rGO over five cycles measured at a scan rate of 1 V/s, from 6.4 mAh/g in the first cycle to 6.3 mAh/g in the fifth cycle. This decline indicates that the electrode material experiences performance degradation as the number of cycles increases. We suspect the decrease in C_{sp} is due to electrode material degradation caused by forming a Solid Electrolyte Interphase (SEI) layer, consistent with previously documented reports [55, 57, 58]. From cycles 2 to 5, the specific capacity stabilizes at 6.3 mAh/g, suggesting that the electrode material begins to achieve cycle stability after the initial cycles.

The curve in Figure 5d shows an increase in C_{sp} (6.08, 9.25, 10.99, 26.82, and 44.15 mAh/g) for NiO-TiO₂-CuO/rGO as the scan rate decreases (1, 0.5, 0.2, 0.1, and 0.05 V/s). This phenomenon is commonly observed and

documented in previous studies [59]. In this study, a high scan rate (1 V/s) results in a lower $C_{\rm sp}$ (6.08 mAh/g) due to the shorter time available for ion intercalation processes. Conversely, a low scan rate (0.05 V/s) yields a higher $C_{\rm sp}$ (44.15 mAh/g) because it allows more time for electrochemical processes. In this case, the obtained capacity is relatively low, which may be attributed to slow reaction kinetics or hindered ion/charge transport within the material.

4. Conclusion

This study successfully developed a NiO-TiO₂-CuO/rGO nanocomposite as an anode material for lithium-ion batteries. Characterization results demonstrate that the material possesses a well-defined crystalline structure with homogeneous particle distribution and an average particle size of 33.25 nm. Electrochemical testing using CV revealed that the nanocomposite exhibits a stable C_{sp} of 6.3 mAh/g after five cycles at a scan rate of 1 V/s, as well as an increased specific capacity of up to 44.15 mAh/g at a scan rate of 0.05 V/s. This enhanced electrochemical performance is attributed to its high cycle stability and improved energy storage capability. Therefore, the NiO-TiO₂-CuO/rGO nanocomposite shows great potential as an efficient and sustainable anode material for lithium-ion battery applications, contributing to the advancement of future energy storage technologies.

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